

# High-throughput Measurements of Viscoelastic Properties Using Surface Indentation

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NCMC-14

November 7<sup>th</sup> 2008

# Buzz Aldrin's Flashlight

- Flashlight used in the Gemini 12 spacewalk, auctioned by Christie's in 1999
- Lens material deformed while in space
  - 1999 NY Times: “Offering persuasive testimony that glass really is a very slow-flowing liquid, not a solid, the flashlight’s thick lens was blistered and warped by the powerful sucking force of the vacuum of space.”
  - Gemini 12 report states it is actually PMMA
- Deformation stresses on the lens from the pressure differential and solar heating



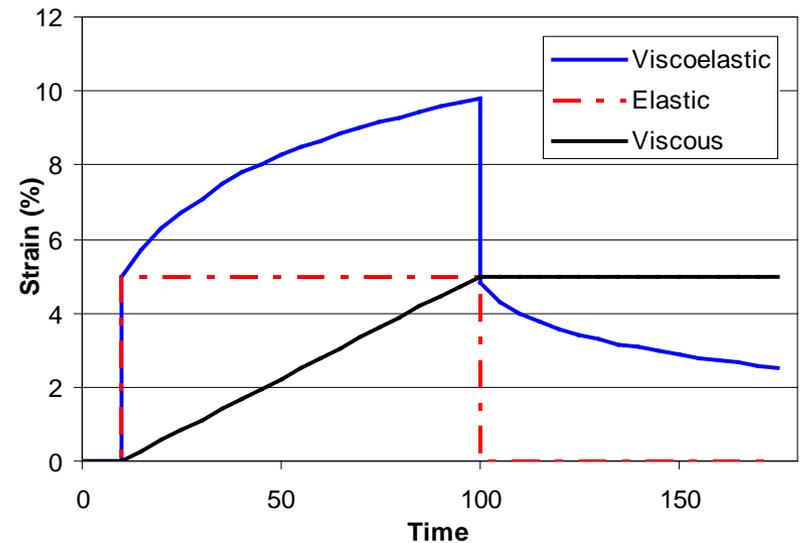
# Nonrecoverable Creep

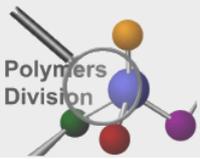
- Jet Engines

- 1300 to 1500 C operating temperature
- Higher  $T_M$  than most superalloys
- Single crystal blades eliminate grain defects and improve resistance to creep

- Structural Metals/Polymers

- Metals and ceramics at high temperatures (30-40% of  $T_M$ ) exhibit creep behavior under load
- Grain defects cause permanent dislocations, leading to failure
- Polymers applications show initial creep – failure occurs under specific conditions



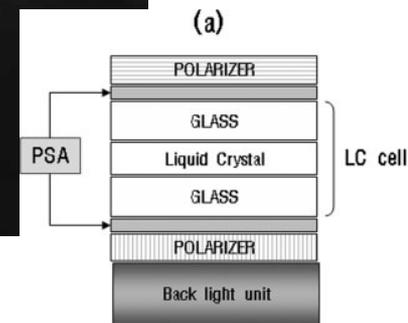


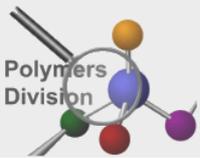
# Polymer Creep

Creep in polymers is a permanent deformation of the polymer network under a static or dynamic load

- PMMA bone cement in hip replacements
- Structural adhesives (permanent bonding)
- Pressure Sensitive Adhesives (temporary/permanent bonding)
- Degradation of polymer networks
- Solvated systems, drug release networks, nanostructured polymers

Creep analysis requires significant experimental time to produce creep compliance curves





# ASTM Creep Standard

- D2990 Standard
- Parallel testing
- Dead weight apparatus
- 10,000 hour duration
  - 416 days

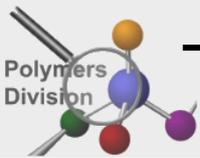


## From D2990

To obtain design data, the test temperatures and environment shall be the same as those of the intended end-use application

Strain load is constant and chosen by the design engineer



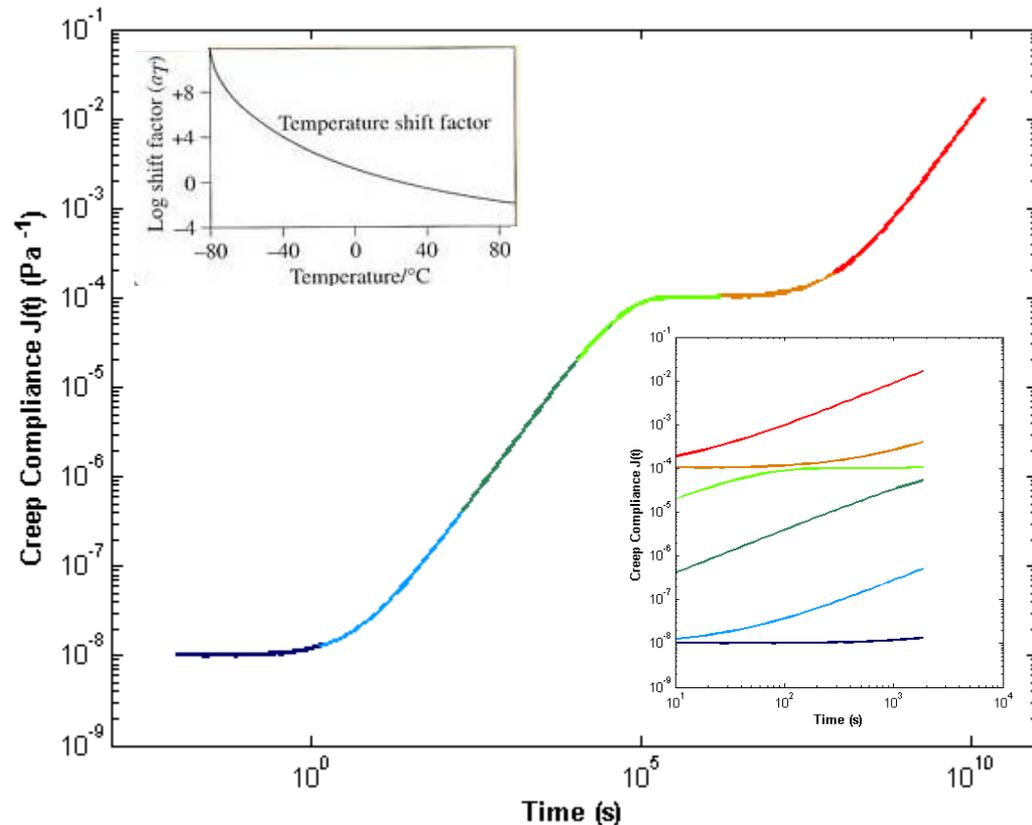


# Time-Temperature Superposition

$$\log a_T = \frac{E_a}{R} \frac{1}{(T - T_0)}$$

$$\log a_T = \frac{C_1(T - T_0)}{C_2 + (T - T_0)}$$

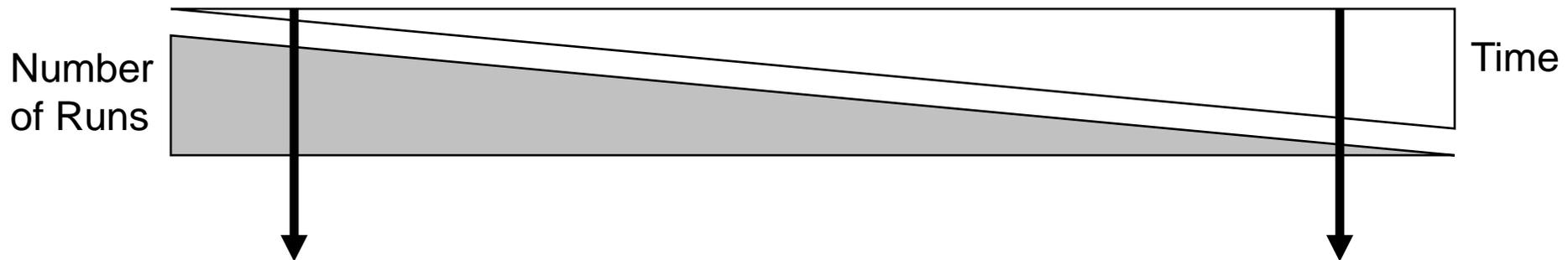
$$\log a_T = a \frac{1 - e^{-(b(T_{ref} - T))^c}}{d + e^{-(b(T_{ref} - T))^c}}$$



- Multiple temperature creep scans are combined into a single master curve (rheometer, DMA)
- WLF: free volume and temperature are linearly related
- Shift factor relationships become more empirical when WLF fails



# Time-Sampling Continuum

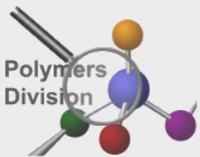


## Time-Temperature Superposition

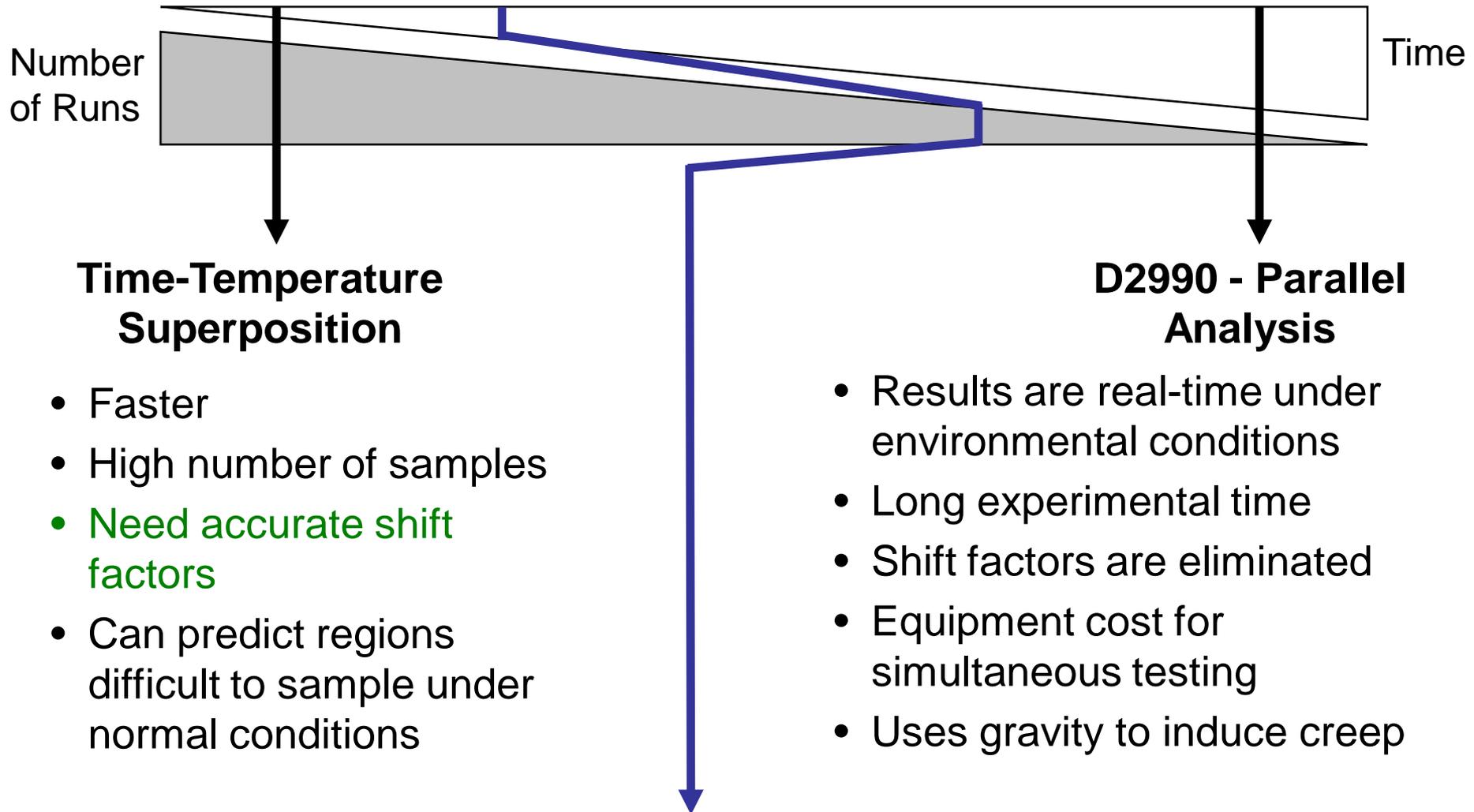
- Faster
- High number of samples
- Need accurate shift factors
- Can predict regions difficult to sample under normal conditions

## D2990 - Parallel Analysis

- Results are real-time with the correct conditions
- Long experimental time
- Shift factors are eliminated
- Equipment cost for simultaneous testing
- Uses gravity to induce creep



# The Middle Ground



Reduce the number of runs required through simultaneous measurements

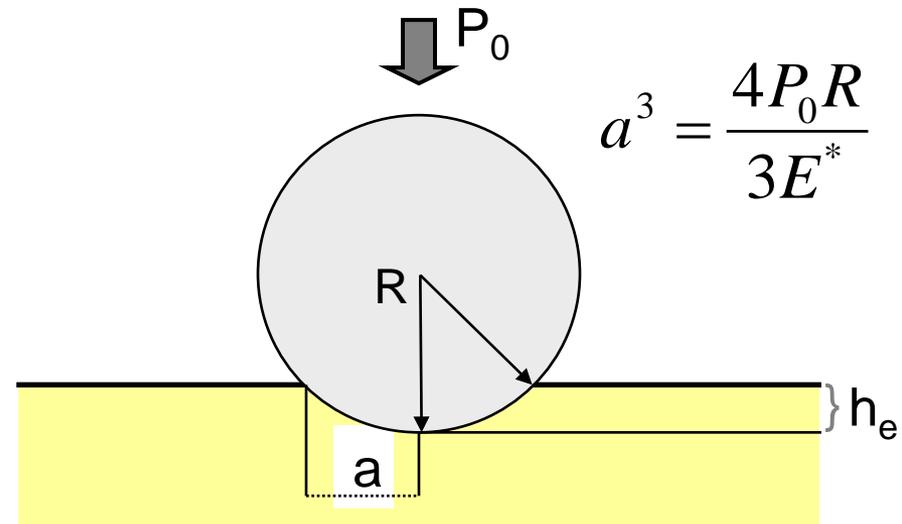
# Contact Area of Sphere During Creep

## Positives

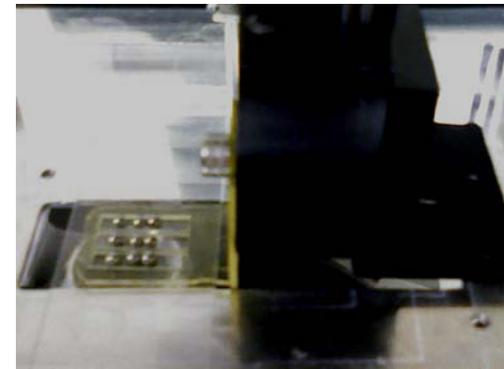
- Known step load from gravity (static force)
- No temperature effect on measurement mechanisms
- Imaging sequence is expandable
- Radius expansion measures a smaller depth indentation
- Spherical nanoindentation

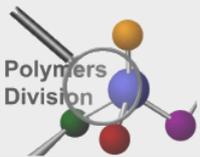
## Negatives

- Measuring visual contact area
- Simple compliance calculations fail for thin films
- Substrate confinement effects depend on adhesion and Poisson's ratio

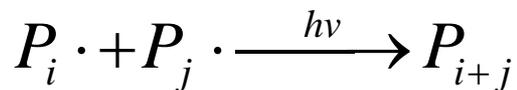


Nanoindentation  $J_C(t) = \frac{8\sqrt{R}h_e^{3/2}(t)}{3(1-\nu)P_0}$

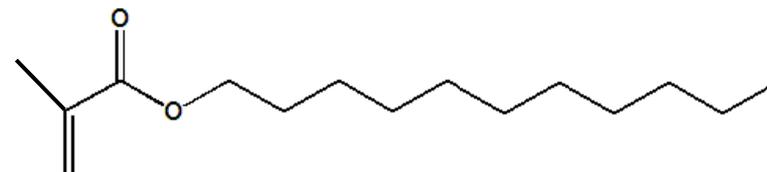




# Photopolymerization

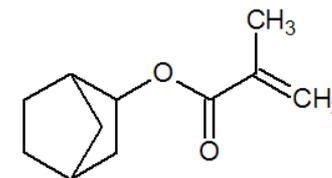


- Solvent-free Processing
- Thick Polymer Samples
- Wide Range of Material Properties
- Broad use in adhesives, automotive materials
- **Broad PDI**
- **Can tailor to produce high rates of creep**



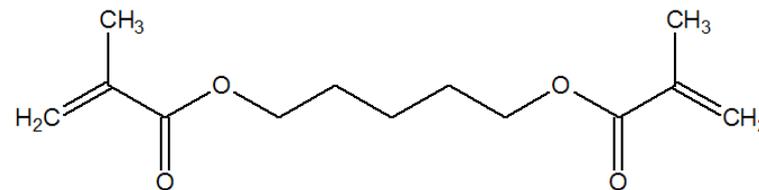
Lauryl Methacrylate (LMA)

*Low  $T_G$*



Isobornyl Methacrylate (IBoMA)

*High  $T_G$*

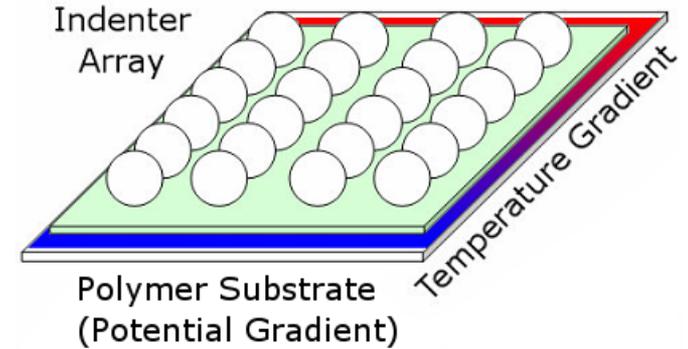


Hexanediol dimethacrylate (HDDMA)

*Crosslinker*

# Multiple Indentations

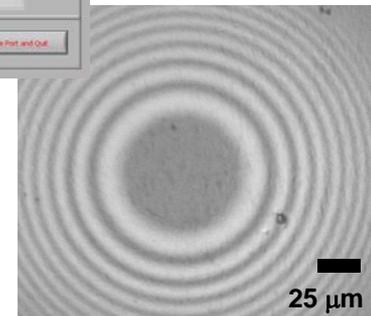
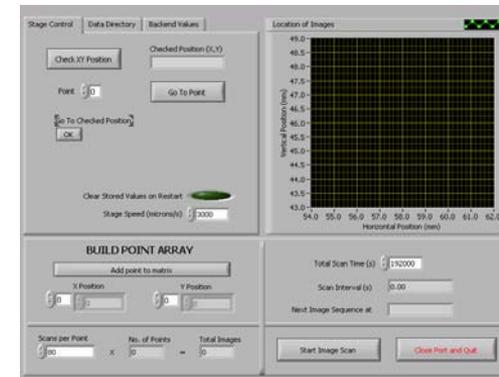
- Temperature gradients for simultaneous time-temperature scans
- Polymer gradients for compositional studies
- Single spheres allow for inexpensive additional measurements on the same substrate

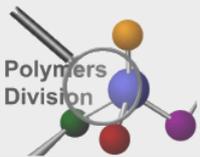


Microscope

## Multiple Sample Process

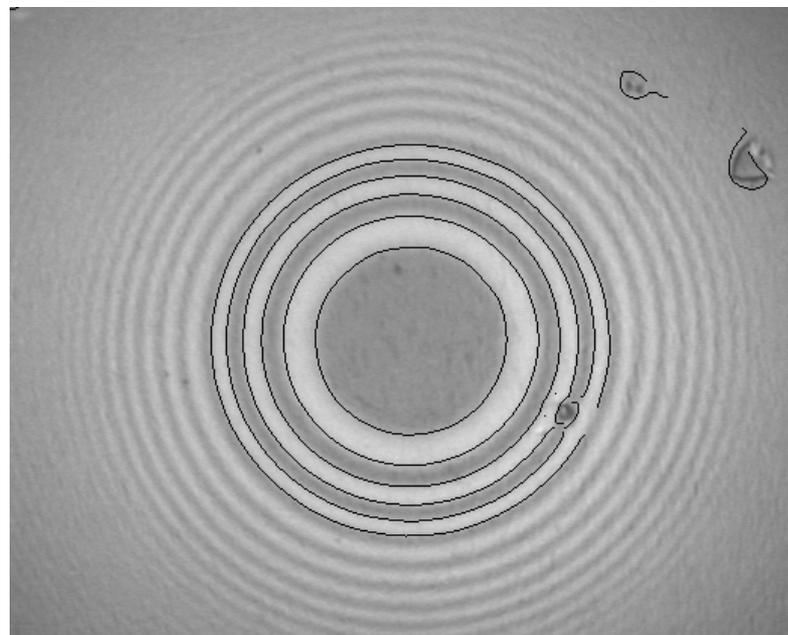
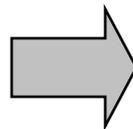
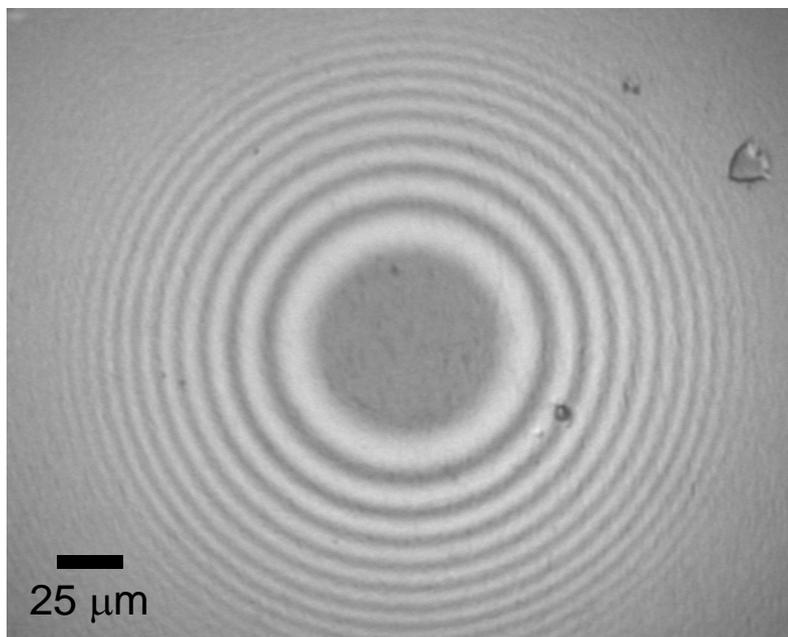
- Raster to each sphere in the array, take an image of the contact point (controlled by LABVIEW)
- Performed for creep greater 100 s because of raster time
- Image processing locates contact area in every image





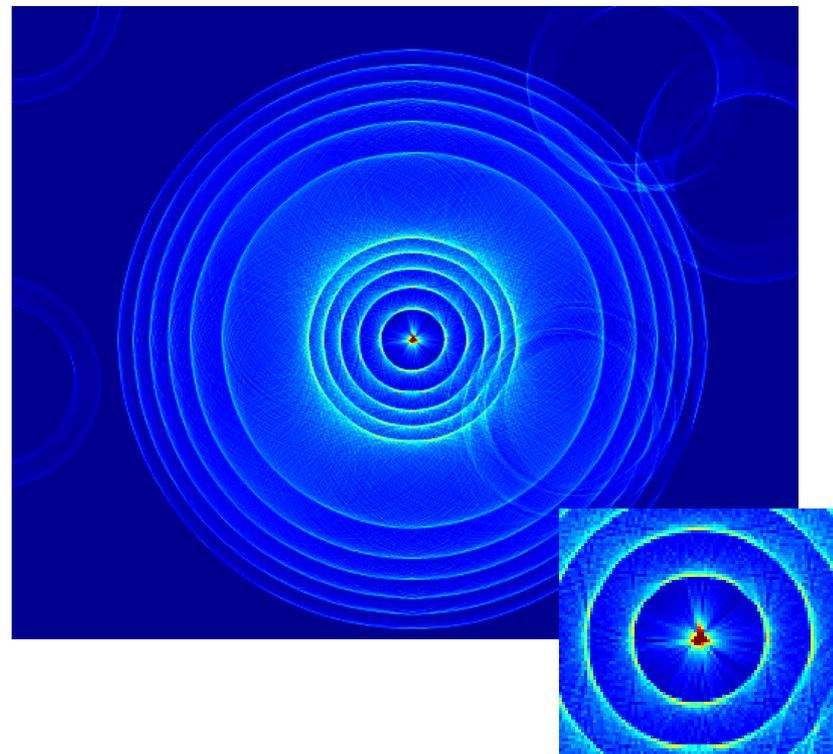
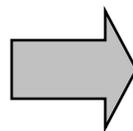
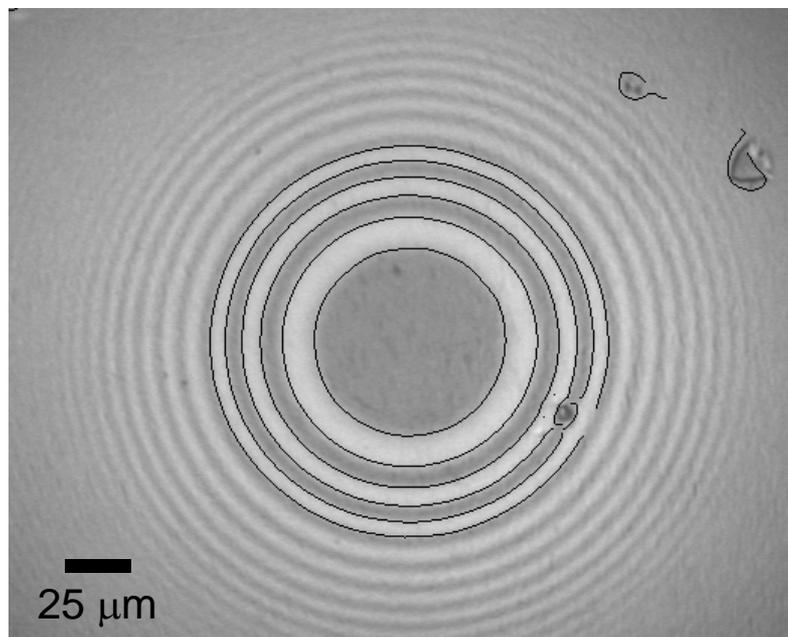
# Processing Results

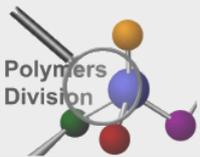
- Find edges of sphere indentation image
- Calculate Hough Transform to find circles
- Report Radius and Circle Position



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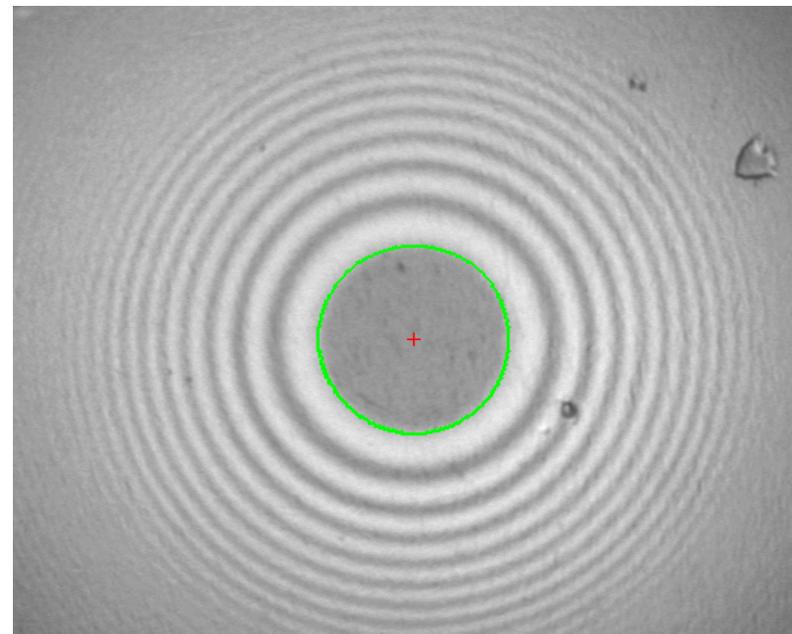
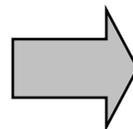
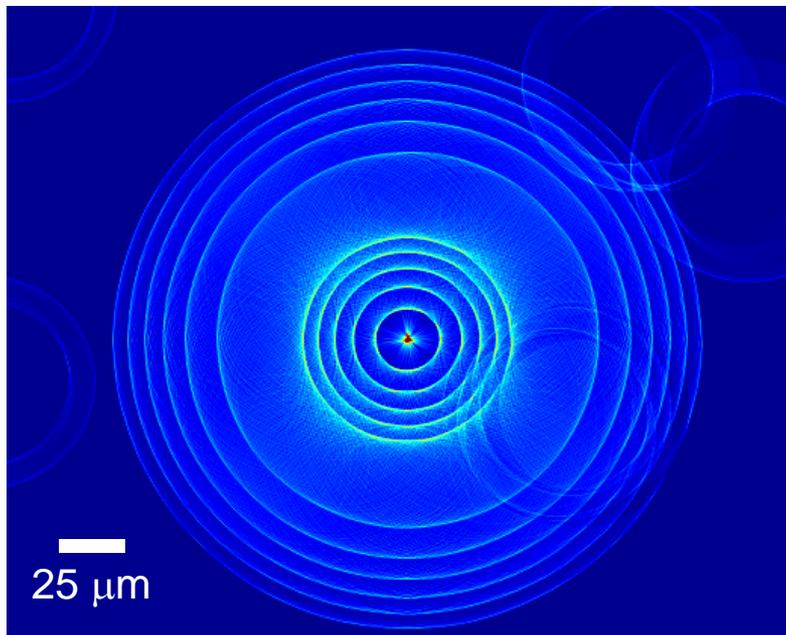
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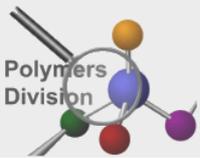




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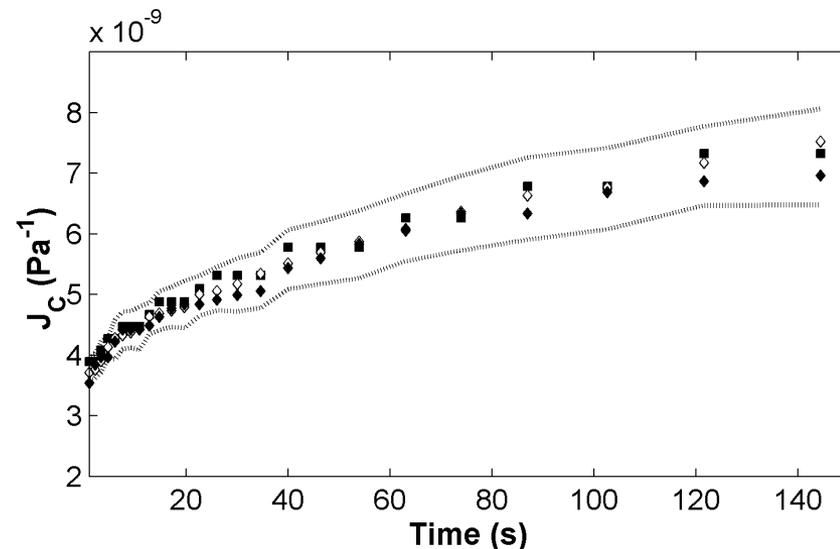
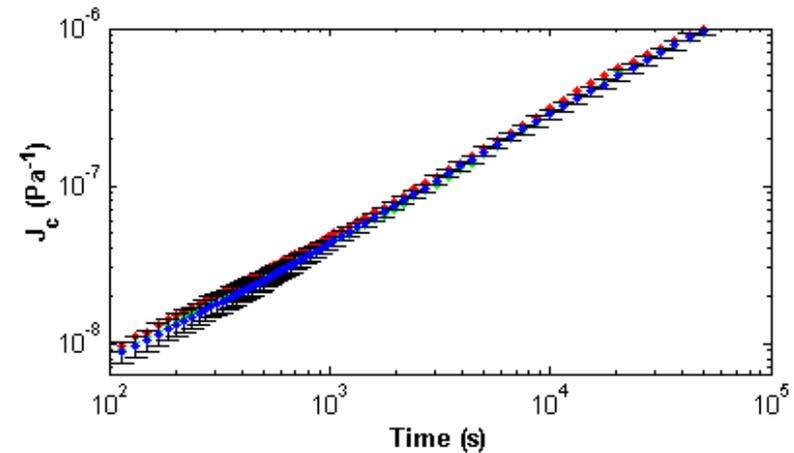
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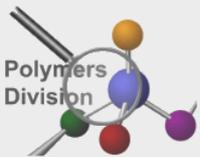




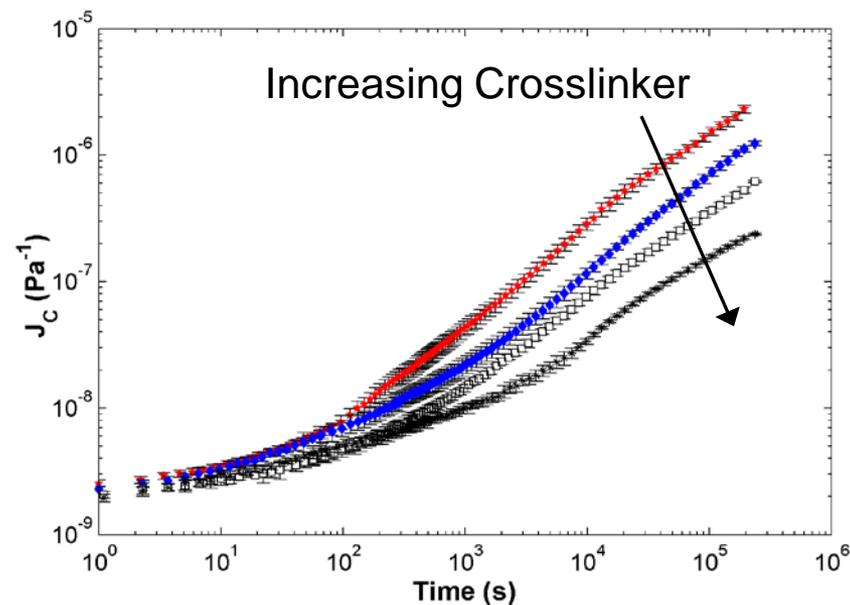
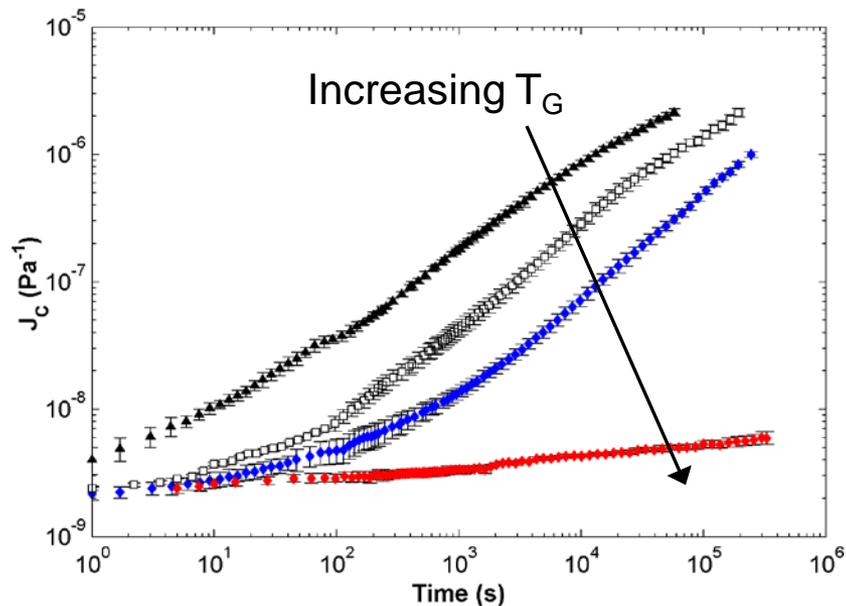
# Limitations and Repeatability

- Statistically equivalent creep compliance over 100000 s (3+ days)
- Load/radius deviation of individual spheres is minuscule in comparison to contact radius error from image analysis
- Small sphere sizes ( $R < 2.5$  mm) produce statistically equivalent results
- Substrate thickness below  $\sim 50$   $\mu\text{m}$  begin to show polymer-substrate effects, correctable for lower film thickness
- Contact effects in highly adhesive system cause non-circular contact areas and higher standard error

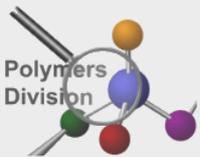




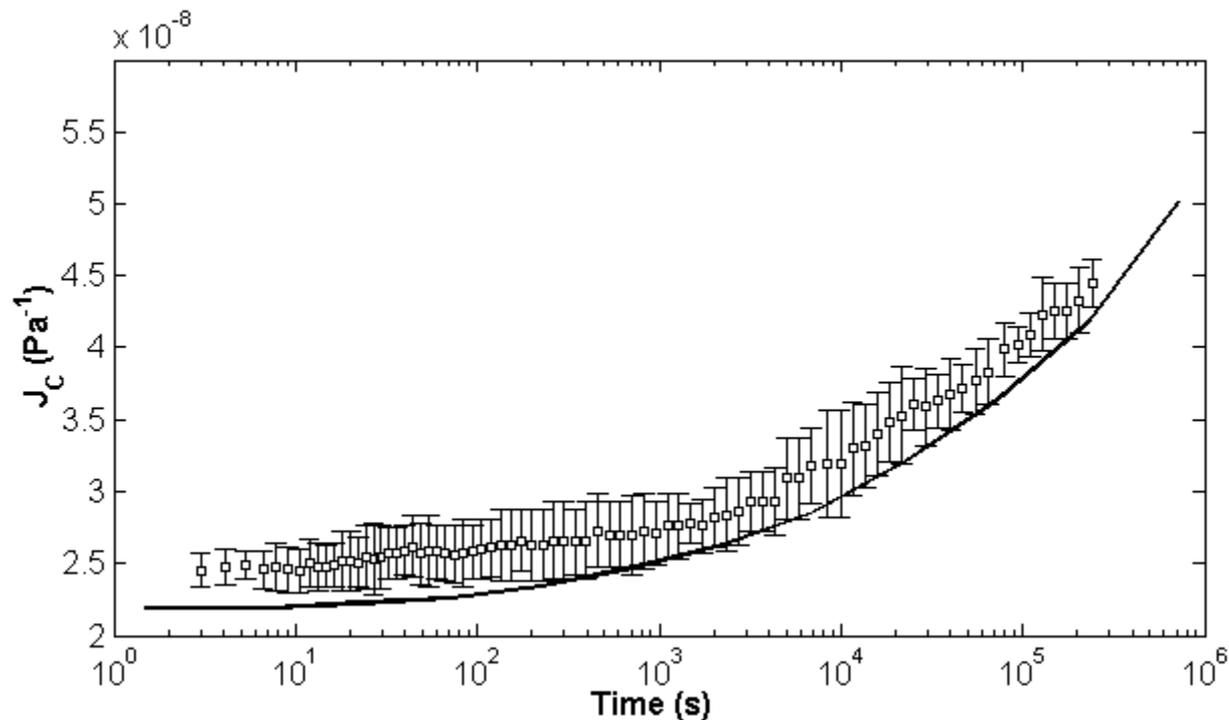
# Compositional Effects



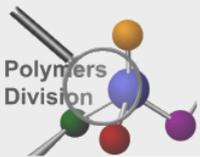
- Higher glass transition temperature polymer systems show reduced creep compliance
- Increasing crosslinking density reduces creep while initial creep compliance remains equivalent



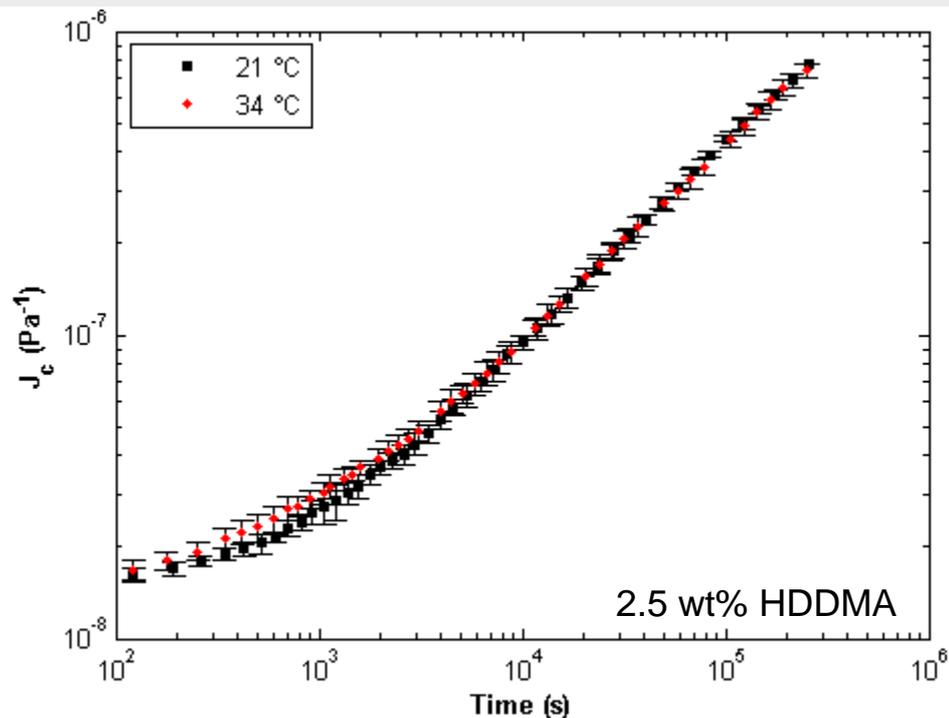
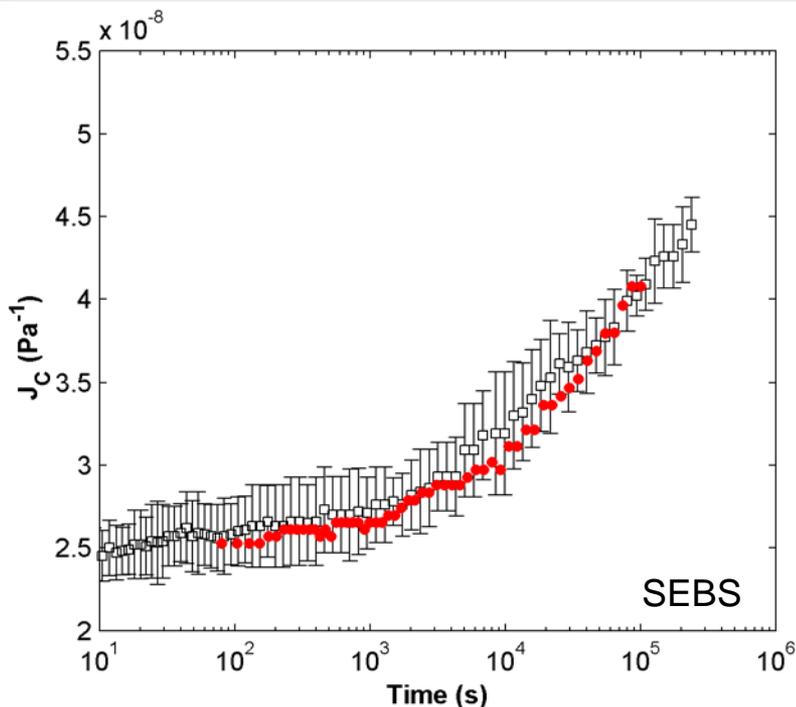
# Comparison to Literature



- Creep compliance measurement slightly higher than reference data at all points
- Different geometry was used with reference curve compiled from time temperature superposition (tensile creep)
- PDMS (Slygard 184) samples also have equivalent compliance values to rheometry and DMA creep tests

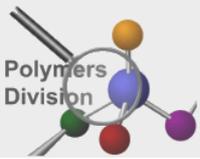


# Time-Temperature Shifting



- SEBS and photopolymer samples show valid time-temperature superposition
- Comparable creep compliance values throughout the entire experiment

Time (s)	21 °C $J_c(t) \text{ (Pa}^{-1}) \times 10^{-8}$	34 °C $J_c(t) \text{ (Pa}^{-1}) \times 10^{-8}$
426	$1.97 \pm 0.09$	$2.24 \pm 0.21$
3975	$5.26 \pm 0.35$	$5.61 \pm 0.39$
11835	$10.48 \pm 0.82$	$10.53 \pm 0.67$
23557	$16.84 \pm 0.97$	$17.17 \pm 1.11$
58026	$30.8 \pm 1.0$	$30 \pm 2.0$
146436	$55.4 \pm 1.9$	$54.4 \pm 3.5$



# Conclusions

- Indentation load is constant and can be measured over the range of creep compliance needed for viscoelastic systems ( $J_C = 10^{-5}$  to  $10^{-9} \text{ Pa}^{-1}$  )
- Samples show consistent trends expected with theory and demonstrate good agreement when compared to DMA or rheometry experiments from literature
- Temperature gradient analysis enables parallel creep compliance measurements for the determination of shift factors of a polymer sample from a single experiment
- Changes in creep compliance for thin films will allow for confinement effects at the polymer-substrate interface to be studied